# Electron Microscopy of Supported Metal Particles I. Behavior of Pt on Titanium Oxide, Aluminum Oxide, Silicon Oxide, and Carbon

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Transmission electron microscopy has been used to investigate the structure and growth characteristics of platinum on thin films of titanium oxide, aluminum oxide, silicon oxide, and carbon. Significant differences in the sintering characteristics of the metal on the various supports was observed after these systems had been treated in hydrogen at temperatures over the range 425 to 1075°K. Sintering occurred on all supports but the metal dispersion was significantly more stable on titanium oxide than the other supports. Examination of platinum on titanium oxide at high resolution showed that the metal particles were predominately hexagonal in outline, of uniform thickness and very thin, indicating a pill-box morphology. Electron diffraction and lattice spacing measurements showed that the titanium oxide support was reduced to  $Ti_4O_7$  in the presence of platinum. In the absence of platinum only the rutile form of TiO<sub>2</sub> was observed. We conclude that platinum catalyzes the reduction of TiO<sub>2</sub> to  $Ti_4O_7$  and deposits as thin pill-box structures on the surface of the  $Ti_4O_7$ . The unique structural features of the type observed for platinum on titanium oxide was not observed with the other supports, and is believed to result from a strong metal-support interaction.

## INTRODUCTION

It has recently been reported that the ability of the Group VIII metal particles to chemisorb hydrogen and carbon monoxide is related to the nature of the support. and the pretreatment of the catalyst (1). For well-dispersed and reduced noble metals on silica, alumina, or carbon, the number of surface metal atoms can be counted by selective chcmisorption of hydrogen or carbon monoxide providing the stoichiometry of the adsorption is

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known. The fraction of exposed metal atoms is then calculated from the hydrogen or carbon monoxide uptakes at saturation coverage and the total number of metal atoms for a unit weight of sample. Decreases in the H/M or CO/M ratios for the aforementioned supports is usually associated with growth of the metal particles following thermal treatment. Titaniasupported metals show markedly different behavior. Following reduction in hydrogen at  $775^{\circ}$ K, H/M and CO/M ratios near zero are found. Such ratios are normally indicative of metal particles much greater than 10 nm in size. X-Ray diffraction and electron microscopy examinations of the titania samples, however, showed no large metal particles indicating that the percentage

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calculated from chemisorption measure- the slide by gradual immersion in water and ments. The suppression of hydrogen and finally mounted on a microscope grid. carbon monoxide chemisorption for tita- Platinum was evaporated at  $10^{-6}$  Torr nium oxide-supported metals was attri-  $(1 \text{ Torr} = 133.3 \text{ N m}^{-2})$  onto these supports buted to an influence of the support on the from a spcctrographically pure platinum chemical and/or structural properties of wire mounted on a tungsten filament. This the metal and has been tcrmcd strong proccdurc produced a platinum film apmetal-support interaction or SMSI  $(1)$ . proximately one atom thick.

In this paper high resolution transmission electron microscopy is used to compare the nature of platinum particles supported on titanium oxide, aluminum oxide, silicon oxide, and carbon films. Films rather than actual catalysts were used to improve the quality and resolution of the micrographs. The objective of these invcstigations was to determine if there are any idcntifiablc fcaturcs which would serve to indicate that metal particles on titanium oxide differ from those on common supports such as alumina, silica, and carbon. Both the variation of metal particle morphology and the dcpcndence of particle size distribution on reduction temperature on all the above mentioned supports were cxamincd.

### EXPERIMENTAL METHODS

Transmission specimens were prepared according to the following proccdurc :

Films of titanium oxide, aluminum oxide, and silicon oxide of about 35 nm thickness were produced by sputtering from the rcspcctivc target materials onto rock salt crystals. The rock salt was subscqucntly dissolved away in water leaving the oxide films floating on the surface. Sections of the films were washed to remove all traces of salt and then picked up on 200-mesh, stainless-steel, electron microscope grids. Carbon films of similar thickness were prepared by vacuum deposition from a carbon arc onto a Pyrex slide, which had been previously coated with a film of detergent.

metal exposed was much larger than that The resulting carbon film was released from

Specimens wcrc subscqucntly rcduccd in a 20% hydrogen/helium mixture under flow conditions for 1 hr at  $425, 825, 975,$ and  $1075^{\circ}$ K. After cooling, they were cxamincd in a Philips EM 300 transmission electron microscope with a resolution better than 0.25 nm. The microscope and its calibration have been described elsewhere  $(2)$ . Particle size distributions were obtained from measurement of over 600 particles after each treatment.

In some cases a shadowing technique was used to detcrminc the height of metal crystallites on the support. This was accomplished by evaporating uranium atoms onto the specimen at an angle of 14° with respect to the plane of the support film. Areas of the specimen shielded from the impinging beam of atoms by surface rclicf features are not coated, and, as a consequence, appear as an enlarged shadow when the specimen is examined in the TEM. Under the conditions used, the height to shadow ratio was 1:4, and thus with the available resolution a crystallite thickness in excess of 2 nm could easily be detected.

## RESULTS AND DISCUSSION

In this section transmission electron micrographs for titanium oxide reduced in hydrogen at various temperatures are first discussed. This is followed by a discussion of the results pertaining to the nature and behavior of platinum particles on titanium oxide, alumina, silica, and carbon. Finally a comparison of the thermal stabilities of metal dispersions on the various supports is made using particle size distribution data.



FIG. 1. (A-D) The appearance of TiO<sub>2</sub> after reduction in H<sub>2</sub> at 425, 825, 975, and 1075°K.

titanium oxide specimens were heated in temperature between 425 and 825°K. At

Effect of Heating Titanium Dioxide in hydrogen are shown in Fig. 1A to D. The Hydrogen original amorphous structure starts to The changes which occurred when blank undergo a change in appearance at some



FIG. 1.-Continued.

975°K it is apparent that this process is lattices overlap and are the result of one of crystallization, as evidenced from periodic matches and mismatches of the the appearance of Moiré fringes  $(3)$  on lattice planes of the overlapping crystals. many areas of the specimen. Moire fringe  $At 1075^\circ K$  the whole surface has undergone patterns can occur where two crystal a reorganization. Platelets up to 60 nm in



FIG. 2. (A) Selected area diffraction pattern of TiO<sub>2</sub> after reduction in H<sub>2</sub> at 425°K. (B) Selected area diffraction pattern of TiO<sub>2</sub> after reduction in  $H<sub>2</sub>$  at 1075°K. (C) Selected area diffraction pattern of Ti<sub>4</sub>O<sub>7</sub> after reducing Pt/TiO<sub>2</sub> in H<sub>2</sub> at 1075°K.

width have grown by transport of material from adjacent areas of the surface, the depleted areas showing up as white zones. Selected area electron diffraction examination revealed that as the temperature was increased, the original diffuse ring pattern (Fig. 2A) gradually changed to a welldefined ring pattern, which had many spots superimposed on the rings (Fig. 2B). This latter pattern was identified as that from the rutile form of  $TiO<sub>2</sub>$ . The mechanism of the observed transformation has been discussed in detail by Shannon  $(4, 5)$  and is thought to be due to the creation of oxygen vacancies when the initial anatase form is heated in hydrogen. No evidence for other

oxides of titanium in the reduced samples was obtained.

# Effect of Heating Supported Platinum Particles in Hydrogen

Examination of the growth characteristics of platinum particles supported on titanium oxide, aluminum oxide, silicon oxide, and carbon showed some profound differences in behavior when these systems were heated in hydrogen. Figures 3 to 6 are transmission electron micrographs taken after heating samples of the four systems to 1075°K in hydrogen for 1 hr. Inspection of these micrographs clearly reveals that there are fundamental differences in behavior of



F1G. 3. Micrograph of Pt on titanium oxide after heating in  $H_2$  to 1075°K for 1 hr.

platinum on the four supports following particles were in general of similar size high temperature reduction. Detailed ex- (3 nm) and appearance on all the supports, amination of many micrographs showed but, as the reduction temperature was

that after heating to only  $425^{\circ}K$ , platinum raised, each system behaved differently.



FIG. 4. Micrograph of Pt on aluminum oxide after heating in  $H_2$  to 1075°K for 1 hr.

pared to the other supports, the platinum- the pure substrate (Fig. 1). However, the

These differences are discussed in detail in titanium oxide system was unique in many the following sections. aspects. In the presence of platinum the (i) Platinum on titanium oxide: evidence titanium oxide appeared to undergo the for metal-support interaction. When com- same crystallization process observed for



FIG. 5. Micrograph of Pt on silicon oxide after heating in Hz to 1075°K for 1 hr.

structure of the titanium oxide after re- 7). Electron diffraction analysis showed the duction at 825°K was markedly different substrate structure in Fig. 7 to be that of when platinum was present. This feature  $Ti<sub>4</sub>O<sub>7</sub>$  (Fig. 2C) rather than the rutile form became more pronounced after reduction of TiOz observed in the absence of platinum. of specimens at higher temperatures (Fig. The crystal structure of  $Ti<sub>4</sub>O<sub>7</sub>$  is triclinic



FIG. 6. Micrograph of Pt on carbon after heating in  $H_2$  to 1075°K for 1 hr.

and has been discussed in detail by Marezio omitted from this paper for the sake of and Dernier  $(6)$ . The formation of Ti<sub>4</sub>O<sub>7</sub> in brevity. The measured spacings are prethe present work was verified by lattice sented in Table 1 along with reported values spacing measurements of the lattice images and corresponding crystal planes (7). The

in Fig. 7 and several other micrographs value of 0.55 nm measured on a number





Measured d (nm)	Literature $d(7)$ . (nm)	[`hkl]
0.62	0.617	002
0.43	0.427	$10\overline{2}$
0.33	0.338	$1\overline{2}0$

TABLE 1 Lattice Spacing Measurements for

of micrographs is most certainly the  $a_0$ unit cell dimension  $(0.56 \text{ nm})$  of Ti<sub>4</sub>O<sub>7</sub>. The formation of  $Ti<sub>4</sub>O<sub>7</sub>$  in the presence of platinum demonstrates that platinum catalyzes the reduction of the support.

The nature and growth characteristics of platinum particles on titanium oxide was also unique among the supports investigated. The platinum particles on titanium oxide increased in size over the temperature range 425 to 825"K, but there did not appear to be any further appreciable increase in size as the temperature was raised to 1075°K. On the other supports particle growth continued up to 1075°K. It was also interesting to find that pretreating the titanium oxide in  $H_2$  at 925°K for 1 hr prior to depositing Pt did not produce any significant difference in the subsequent growth characteristics of Pt compared to that on untreated amorphous titanium oxide. The 925°K hydrogen pretreatment caused the titanium oxide substrate to crystallize as was shown in Fig. 1. Since the platinum growth characteristics were the same starting with amorphous or crystalline titanium oxide, the crystallization process has little or no influence on platinum particle growth. In both cases the final structure of the substrate was that of  $Ti<sub>4</sub>O<sub>7</sub>$ .

Another unique aspect of the platinumtitanium oxide system was the observation that the electron scattering density of the platinum particles remained virtually unchanged over the temperature range 825- 1075°K indicating that particle growth was not occurring by an increase in particle height. Examination of these specimens at very high magnification  $(\sim 2.5)$  $\times$  10<sup>6</sup>) revealed several other features of the platinum particles. They were found to be predominately hexagonal in outline and not only were they very thin, but the thickness was uniform across the particle indicating the morphology to be that of a pill-box rather than hemispherical. The failure to find any trace of a shadow after evaporation of uranium places an upper limit on the particle thickness of  $\sim$ 2 nm, i.e., a few atomic layers.

Additional evidence for the thin pill-box structure of the platinum particles on titanium oxide was obtained from further inspection of the high magnification micrograph (Fig. 7). It can be seen that lattice images of the  $Ti<sub>4</sub>O<sub>7</sub>$  substrate are clearly visible through most of the platinum particles. Moreover, the intensity of such fringes appears uniform across the complete width of a given particle confirming the uniform thickness of a particle, i.e., pill-box morphology. Defocusing experiments verified that the platinum particles were on top of the substrate. Similar observations were made by Prestridge et al. (8) in their studies of metal clusters on silica, which led them to conclude that ruthenium and ruthenium-copper clusters were in the form of rafts, one and two atomic layers thick, respectively. The present results do not, allow us to make the same definitive claims, but we do believe that platinum particles are present as extremely thin flat structures.

The observations that the support is reduced to  $Ti<sub>4</sub>O<sub>7</sub>$  only in the presence of platinum and that platinum is present as thin flat particles, even at 1075°K when the energetically preferred shape would be hemispherical, suggest that a structural relationship may exist between the metal and the support. Two possibilities, epitaxy and pseudomorphism, merit consideration in this regard.

For the case of epitaxy, the metal

particles retain their normal lattice spacing Although the results of this study do not<br>but assume the same orientation as that allow us to draw any definite conclusions layers in thickness (10, 11). In the case of theless point to an unusual structural pseudomorphism, the lattice spacing of relationship between the metal and oxide. the deposit material is identical to that of Furthermore, the results are in accord with the substrate  $(12)$ . It appears unlikely that the chemisorption studies of Tauster *et al.* a pseudomorphic deposit would be more  $(1)$  which demonstrated that the suppresthan one atomic layer thick if there were sion of  $H_2$  and CO adsorption on Pt-titanan appreciable difference between the ium oxide are indicative of strong chemical normal lattice spacing of the deposit and perturbations on the platinum due to that of the substrate. In principle, electron metal-support interaction. This result prodiffraction patterns showing the relative vides indirect evidence for a strong chemical orientations of the deposit and substrate bonding between the metal and support can establish if epitaxy exists. The plat- that could give rise to pseudomorphism. inum-titanium oxide samples, however, The fact that the platinum is deposited on showed no electron diffraction pattern for a reduced form of titanium oxide,  $Ti<sub>4</sub>O<sub>7</sub>$ , platinum, so we are in no position to make suggests that platinum atoms may form any strong claims regarding the nature chemical bonds with  $Ti<sup>3+</sup>$  cations. Bonding of the interaction in this system. Additional between platinum cations and  $Ti<sup>4+</sup>$  cations experimental data from techniques such is known to occur in solid-state compounds, as EXAFS which could provide information namely the hexagonal barium titanates on the local environment of platinum or  $(13)$ . For the Pt-titanium oxide system titanium would clearly be valuable in bonding between platinum and titanium

but assume the same orientation as that allow us to draw any definite conclusions of the substrate surface  $(9)$ . Crystallites regarding the precise nature of the interregarding the precise nature of the interexhibiting epitaxy may be many atomic action between Pt and  $Ti<sub>4</sub>O<sub>7</sub>$ , they neverfurther understanding the  $Pt-Ti_4O_7$  system. would appear to involve a platinum atom



FIG. 8. Variation of mean platinum particle size with reduction temperature on the four substrates.

cations so that bonds to an aggregate of particle agglomeration. At  $1075\,\mathrm{K}$  a draplatinum atoms can form. Recent theoret- matic change in appearance of the surface ical calculations in this laboratory by the is seen ; small particles have disappeared x-alpha scattered wave method indicate and large dense areas have formed. This that covalent bonding can indeed occur behavior was observed in dynamic studies between a platinum atom and titanium of the interaction of  $Pt/Al_2O_3$  with  $O_2$ cations at the surface of titanium oxide;  $(15)$ . Recently Sprys and Mencik  $(16)$ in addition, an ionic contribution involving found the same effect when  $Pt/A<sub>12</sub>O<sub>3</sub>$  $Ti<sup>+3</sup>$  ions is also under consideration  $(14)$ . specimens were subjected to intense elec-

rather than a cation because Tauster *et al.* (ii) Platinum on aluminum oxide, silicon (1) showed that platinum was reduced to *oxide*, and *carbon*. In the platinum on (1) showed that platinum was reduced to *oxide*, and carbon. In the platinum on the metal at low temperatures  $(475^{\circ}K)$  aluminum oxide system expected particle the metal at low temperatures  $(475^{\circ}K)$  aluminum oxide system expected particle and only interacted strongly with the growth is observed as the temperature is growth is observed as the temperature is support on further reduction at  $775^{\circ}K$ . increased from 425 to 975 $^{\circ}K$ , but at the For platinum at the surface of titanium upper temperature it is evident that the oxide, reduction of the support to a lower number of isolated particles present has oxide may be necessary to provide a decreased to a far greater extent than can sufficient concentration of exposed titanium be accounted for purely on the basis of



FIG. 9. Variation of PSD with reduction temperature for Pt on titanium oxide.

tron beams within the electron microscope and suggested that it was due to the formation of a Pt-Al alloy. In the present investigation this condition was carefully avoided by the use of low beam intensities and we saw no enhancement of the interaction during the microscopy examination. Clearly the nature of the gas environment does not appear to have any influence over this reaction and raises serious questions as to the formation of a  $Pt-Al$  alloy. Although the Pt particles have a faceted outline at  $825^\circ K$ , it is not possible to discern any one particular preferred shape.

A common feature encountered in both the Pt on titanium oxide and Pt on aluminum oxide systems is that all the metal particles remain as discrete entities, there being no tendency for particles to stick together in the form of chains. This observation suggests that once formed, the particles remain static on the support surface and almost certainly grow by migration of atomic species from one crystallite to another.

Examination of the micrographs of the Pt on silicon oxide system indicates that particle growth is quite rapid and that there is extensive agglomeration as the reduction temperature is increased. At the intermediate temperatures, 775 to  $975^{\circ}$ K, the metal takes the form of clongatcd "puddles" and there is a great deal of evidence for neck formation between adjacent islands of platinum indicative of liquid-like coalescence.

The behavior of platinum on carbon is similar in some respects to that of the Pt on silicon oxide system in that particles tend to bc relatively globular in outline and there is cvidcncc of liquid-like character. The picture is further complicated by the fact that platinum has the ability to catalyze the hydrogenation of the carbon support at temperatures above 775°K  $(17)$ . The results of this action can



FIG. 10. Variation of PSD with reduction temperature for Pt on aluminum oxide.



**FIG. 11.** Variation of **PSD** with reduction temperature for **Pt** on silicon oxide.

be seen in Fig. 6, where channels have been created in the wake of almost all the platinum particles. These channels have been caused by the preferential removal of carbon aroms to produce methane at the metal-carbon interface and as a consequence, particles are drawn across the surface removing carbon in their path. This type of behavior would produce the same effects in the carbon as observed during catalytic oxidation of carbon by platinum particles **(18).** Because motion is imparted to the particles by this reaction, there is a high probability of particle collisions leading to coalescence, and a resulting enhancement in the particle growth rate.

(iii) *Thermal stability of metal dispersions* 

*for the various substrates.* Quantitative determinations of the mean particle size dependence on temperature are presented in Fig. 8, where it can be seen that Pt particles supported on titanium oxide are at least a factor of 2 smaller than on the other supports at the higher temperature range. A detailed analysis of particle size distributions at various temperatures (Figs. g-12) provides a more enlightening picture of the particle growth characteristics in these systems and reveals significant differences in the behavior of Pt on titanium oxide in comparison to the other supports. It is evident that the distribution from Pt on titanium oxide is narrower than for the other substrates at all temperatures and shows little change in magnitude over



FIG. 12. Variation of PSD with reduction temperature for Pt on carbon.

the range  $S25$  to  $1075^{\circ}$ K with the exception of the tendency to form a splinter group of small particles. The distribution from Pt on aluminum oxide gets progressively broader with increasing temperature, a trend which is much more pronounced in the Pt on silicon oxide and Pt on carbon systems.

The particle size distributions as a function of reduction temperature give further evidence of strong metal-support interaction for Pt on titanium oxide. Regardless of whether one postulates an atomic migration mechanism or a particle migration mechanism for agglomeration, we expect increased metal-support bonding to inhibit sintering as is observed with Pt on titanium oxide.

# $\mathcal{B}$ UMMARI $\mathbf{I}$

High resolution transmission electron microscopy of Pt on titanium oxide, aluminum oxide, silicon oxide, and carbon shows that following reduction at low temperature  $(425\text{°K})$  the platinum particles are similar in size and appearance on all the substrates. Following reduction at  $825^{\circ}$ K or higher, however, Pt on titanium oxide assumes a pill-box structure of hexagonal shape and the substrate is reduced to  $Ti<sub>4</sub>O<sub>7</sub>$ . In the absence of Pt the titanium oxide is not converted to

 $Ti<sub>4</sub>O<sub>7</sub>$ . Detailed analysis of high resolution micrographs suggest that the Pt forms a structural relationship with the  $Ti<sub>4</sub>O<sub>7</sub>$ . Such a relationship might require the formation of chemical bonds between the metal and the support. The marked alteration of the Hz and CO chemisorption properties of Pt on titanium oxide reported recently by Tauster *et al. (I)* provides evidence that such metal-support bonding exists in this system. No unique structural features were observed with the other supports, aluminum oxide, silicon oxide, and carbon.

Particle size distribution data as a function of thermal treatment in hydrogen demonstrated that the order of stability of the platinum dispersions on the substrates was titanium oxide > aluminum oxide  $>$  silicon oxide  $\approx$  carbon. This again is indicative of a metal-support interaction in the Pt-titanium oxide system.

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